

Thermoelectric Properties of Self Assemble $\text{TiO}_2/\text{SnO}_2$ Nanocomposites

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Recent advances in improving efficiency of thermoelectric materials are linked to nanotechnology. Thermodynamically driven spinodal decomposition was utilized to synthesize bulk nanocomposites. $\text{TiO}_2/\text{SnO}_2$ system exhibits a large spinodal region, ranging from 15 to 85 mole % TiO_2 . The phase separated microstructures are stable up to 1400 °C. Semiconducting $\text{TiO}_2/\text{SnO}_2$ powders were synthesized by solid state reaction between TiO_2 and SnO_2 . High density samples were fabricated by pressureless sintering. Self assemble nanocomposites were achieved by annealing at 1000 to 1350 °C. X-ray diffraction reveal phase separation of $(\text{Ti}_x\text{Sn}_{1-x})\text{O}_2$ type phases. The $\text{TiO}_2/\text{SnO}_2$ nanocomposites exhibit n-type behavior; a power factor of $70 \mu\text{W}/\text{mK}^2$ at 1000 °C has been achieved with penta-valent doping. Seebeck, thermal conductivity, electrical resistivity and microstructure will be discussed in relation to composition and doping.



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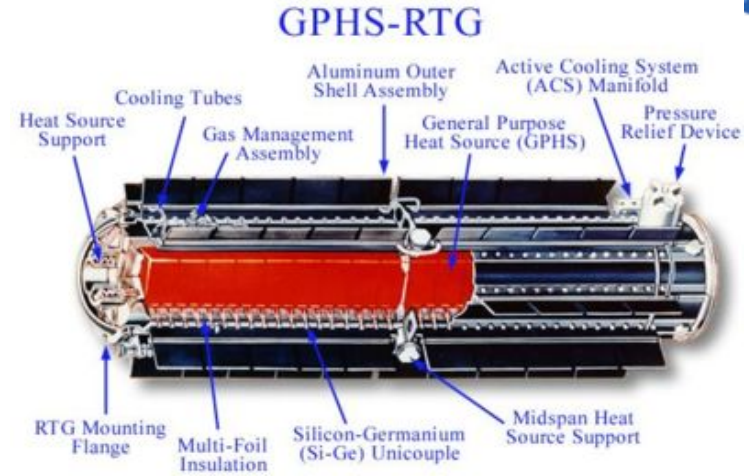
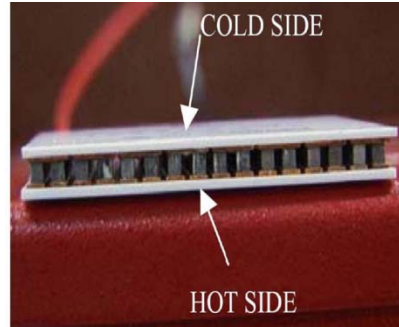
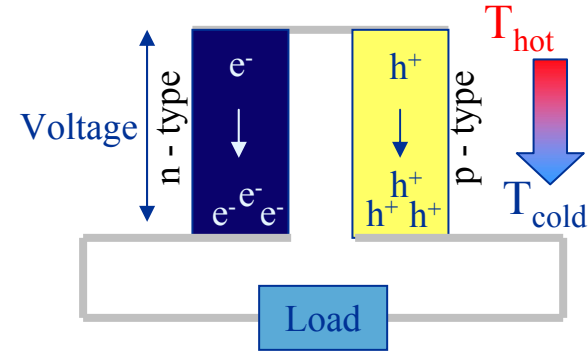
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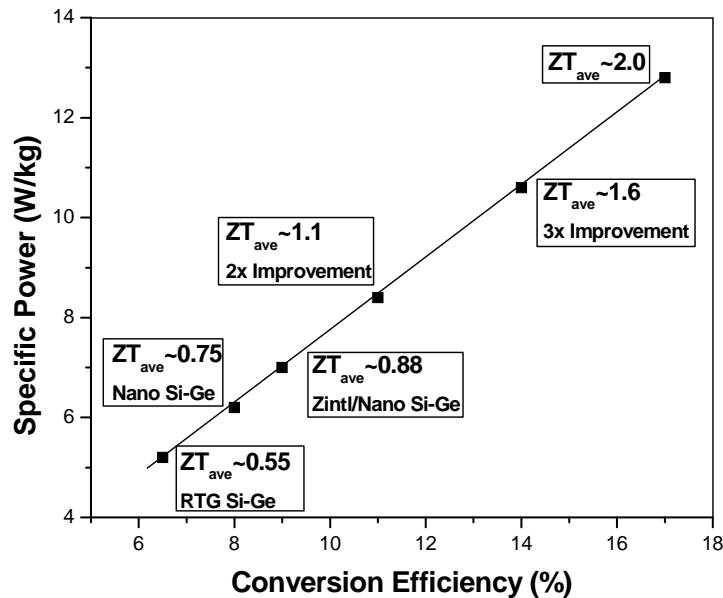
Program Support: NASA Radioisotope Power Systems

Heat to Electric Power Generation



Objective: High Conversion Efficiency
 ● Reduces Mass, Volume & Cost

Space Power Generation



Waste Heat to Power

- Waste Heat is one of our most under utilized energy resources
- U.S.-energy consumption ~29 tera-kWh (10^{12})
Barrels of Oil – 170 giga-barrels (10^9)
- World-energy consumption ~120 tera- kWh (10^{12})
- 20-65 percent is lost in the form of heat
- Maximizes efficiency
- Reduces CO₂ emission

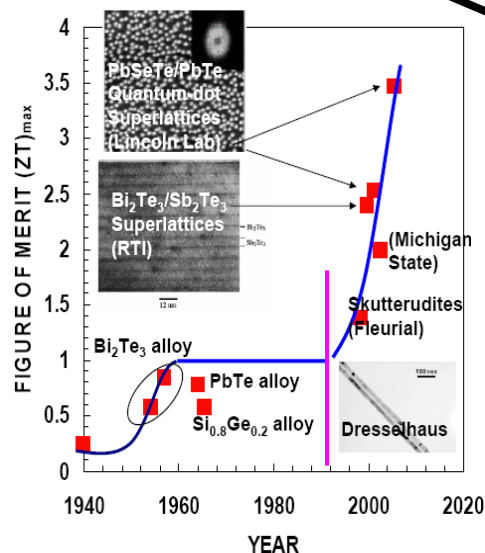
Figure of Merit

$$ZT = \frac{S^2 \sigma}{\kappa} T$$

S - Seebeck coefficient
 σ - electrical conductivity
 κ - thermal conductivity

Efficiency

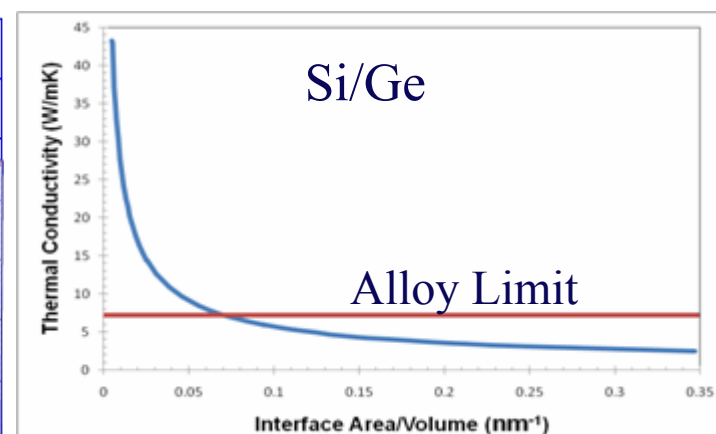
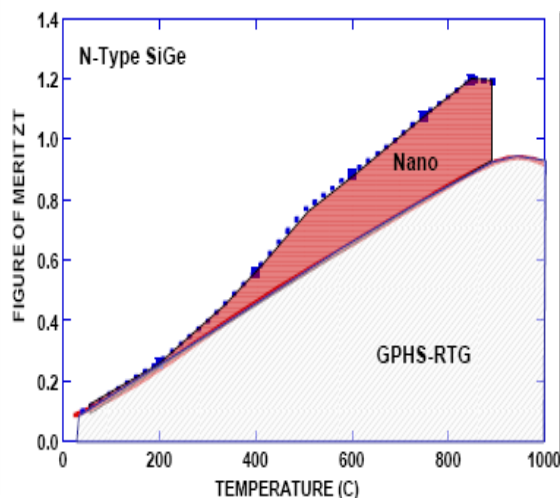
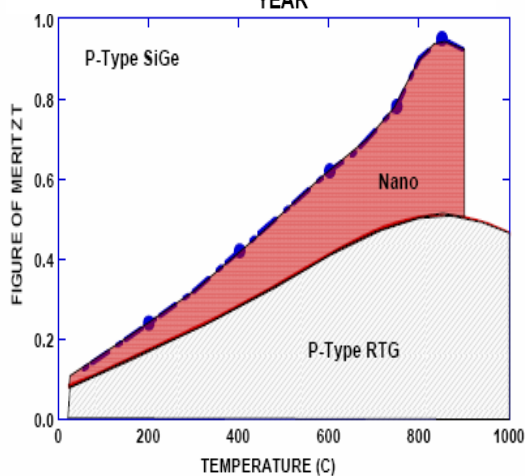
$$\eta_{\max} = \frac{\Delta T}{T_{\text{hot}}} \frac{\sqrt{1 + ZT} - 1}{\sqrt{1 + ZT} + T_{\text{cold}}/T_{\text{hot}}}$$



Phonon Scattering:

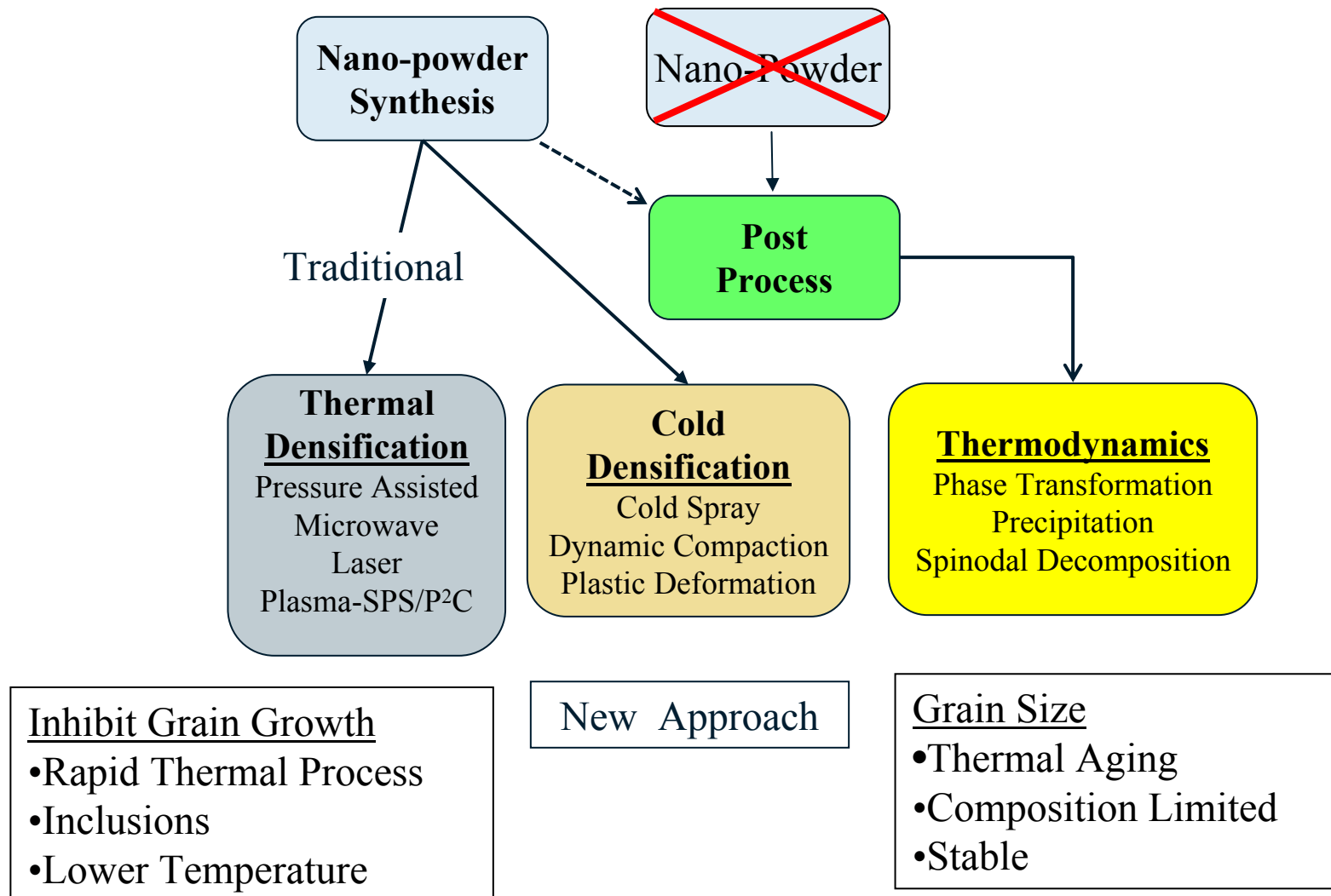
- Atom disorder
- Alloying
- Anharmonic vibrations
- Supperlattices
- Crystal Structures
- Nano-technology

Fleural/Chen – JPL/MIT



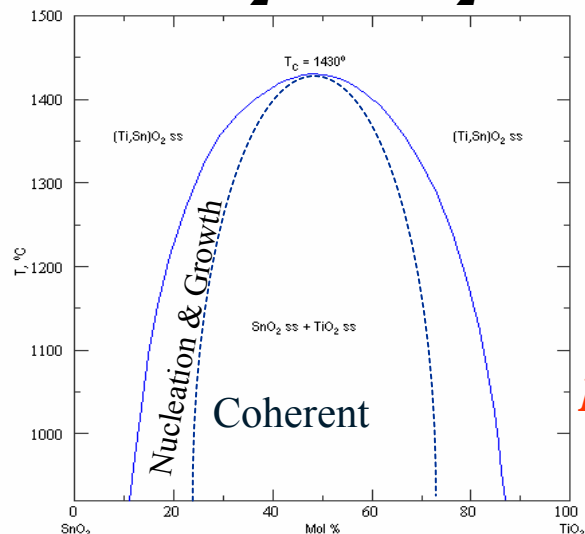
Fabrication of Nanostructure Solids

Goal: Preservation of the nanostructure during fabrication.



Spinodal Decomposition

TiO₂ – SnO₂



Desired Features

- ~50 nm grains
- High Temperature
- Wide Composition
- Large Δ Mass

Transparent Conducting Oxides

Insulator/Semiconductor/Conductor

- Large Bandgap 2.4-3.8 eV
- N-type – Degenerate Semiconductor

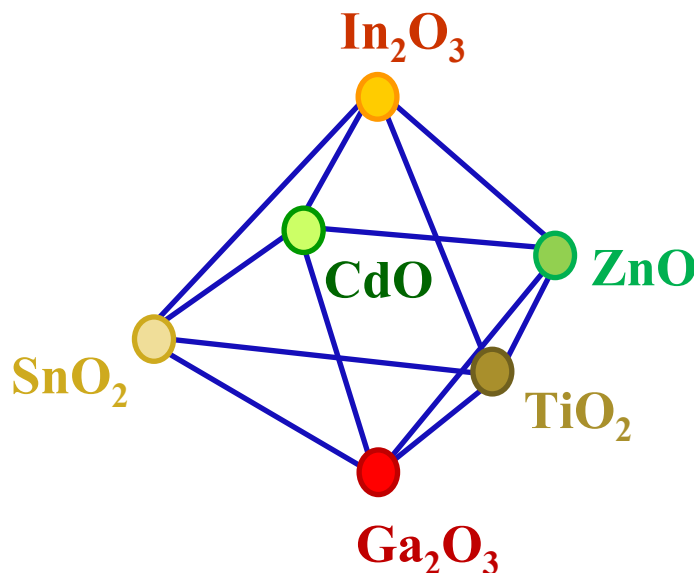
Electrical Conductivity

TCO	σ (S/m) @ RT
ITO	8×10^5
In ₂ O ₃	1×10^6
SnO ₂	2.5×10^5
ZnO	8.3×10^5
ZnO:Al	7.7×10^4
CdSnO ₂	7.7×10^5
CdO:In	1.7×10^6



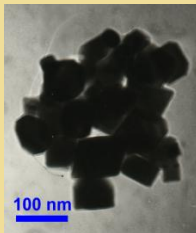
Fig. 10. TEM image of (Ti_{0.5}/Sn_{0.5})O₂ ceramics annealed for 48 h.

Shultz & Stubican, JACS, 53, 1970

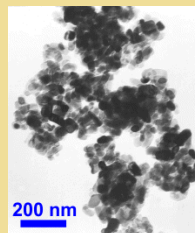


ZnO:Al
ZT~0.6 @ 1000 °C

SnO_2
Purity: 99.9%
APS: 50 nm
SSA: $14.2 \text{ m}^2/\text{g}$



TiO_2 Rutile
Purity: 99.99 %
APS: 20 nm,
SSA: $> 30 \text{ m}^2/\text{g}$



Dopants
 CoO , MnO_2
 Ta_2O_5 , In_2O_3

$\text{TiO}_2/\text{SnO}_2$
50/50 mol %
75/25 mol %
25/75 mol %

Powder
Mixing

Compaction
Die Press

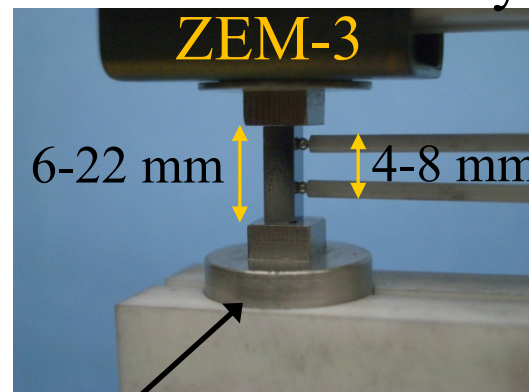
Reactive Sintering
1250-1550 °C

Anneal
72 Hrs

Thermal Conductivity

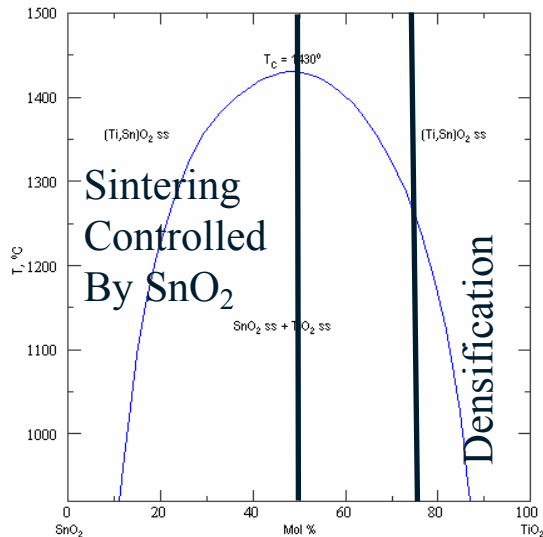
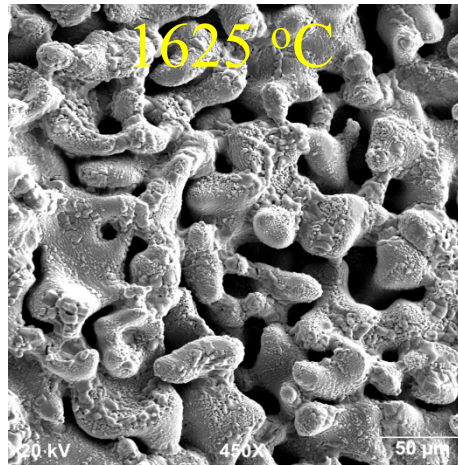
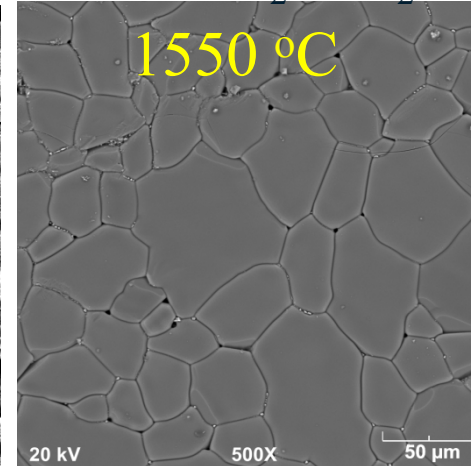
- Laser Flash Method- Thermal Diffusivity
- Standard
- Specific Heat- C_p - Laser Flash
- Thermal Conductivity ($K = \alpha \rho C_p$)

Seebeck/Resistivity



ΔT 0-50 °C/Furnace RT-1000 °C

Sintering

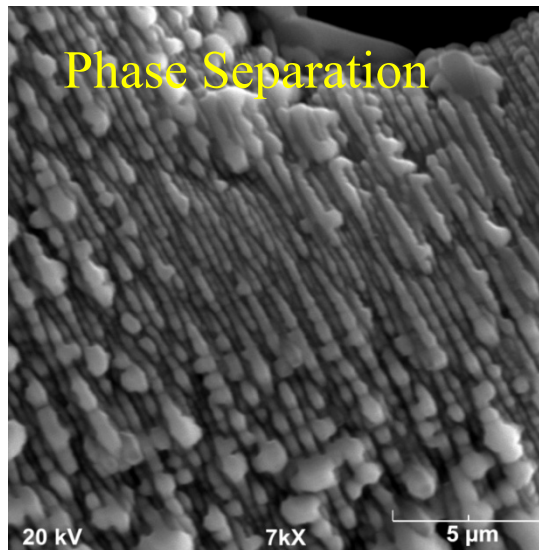
50/50 TiO₂/SnO₂75/25 TiO₂/SnO₂

SnO₂ Sintering-Inhibited

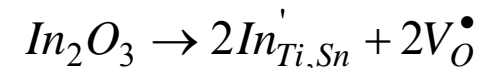
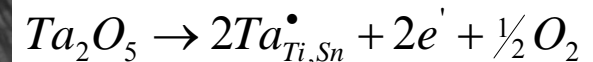
- Surface Diffusion < 1100 °C
 - Evaporation > 1100 °C
- $$\text{SnO}_2 \rightarrow \text{SnO} + \frac{1}{2}\text{O}_2(\text{g})$$

Sintering Aids-SnO₂

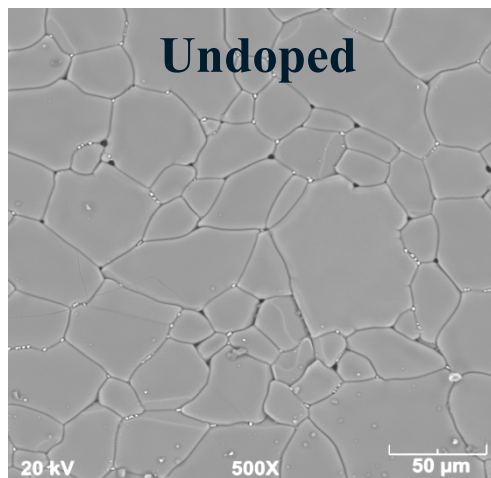
- MnO, CoO, CuO, ZnO

50/50 TiO₂/SnO₂

Ta₂O₅ & In₂O₃
Ineffective Sintering Aids

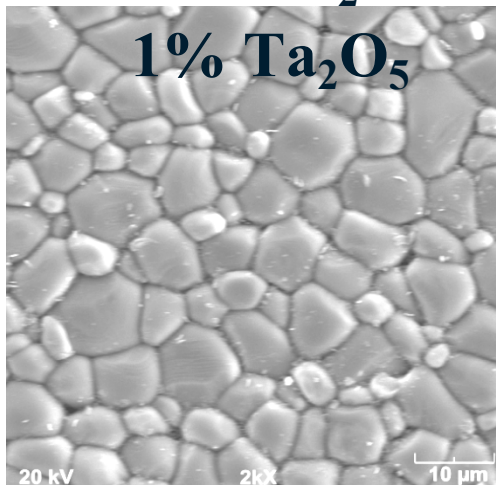


75/25 $\text{TiO}_2/\text{SnO}_2$



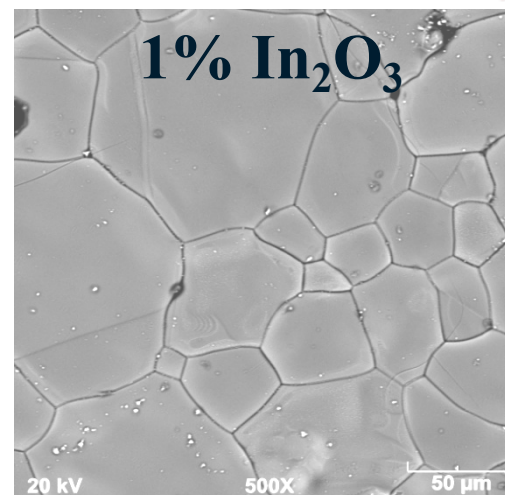
XRD-Phases

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 Reduced – TiO_2 , Rutile
 $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$



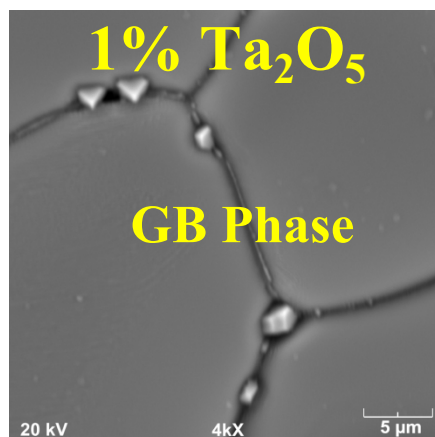
XRD-Phases

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 Annealed – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 1250 °C
 Reduced – TiO_2 , Rutile
 $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$



XRD-Phases

Sintered – TiO_2 , Rutile
 SnO_2 , In_2O_3
 Annealed – TiO_2 , Rutile
 1250 °C SnO_2 , In_2O_3



Phase Separation

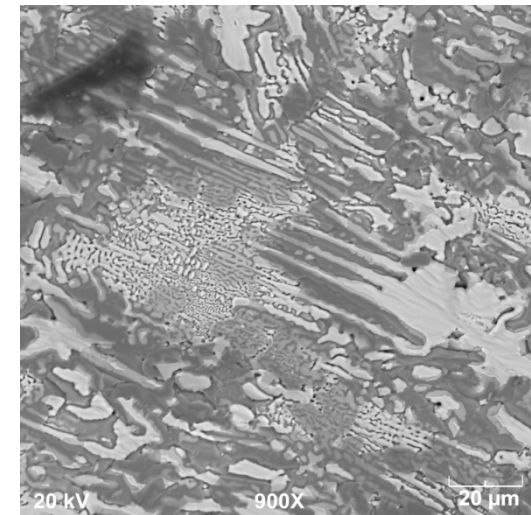
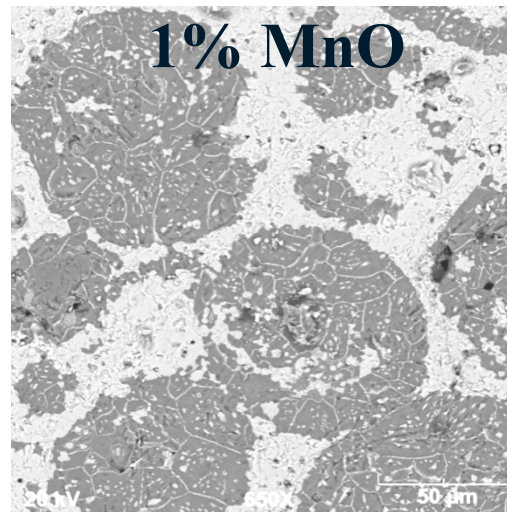
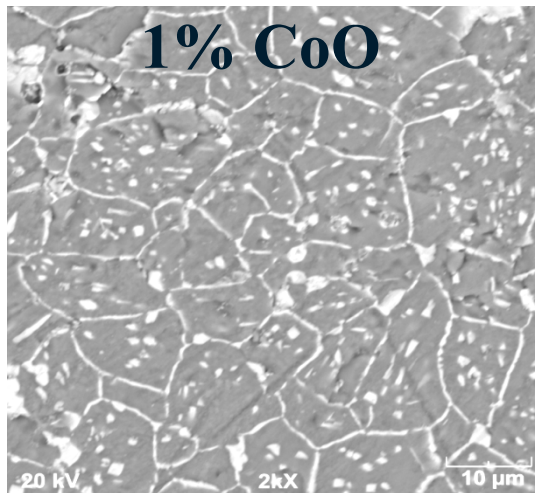
1% CoO XRD

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 $(\text{Ti}_{0.2}\text{Sn}_{0.8})\text{O}_2$
 Annealed – $(\text{Ti}_{0.9}\text{Sn}_{0.1})\text{O}_2$
 1000 °C $(\text{Ti}_{0.1}\text{Sn}_{0.9})\text{O}_2$

1% MnO XRD

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 $(\text{Ti}_{0.2}\text{Sn}_{0.8})\text{O}_2$
 Annealed – $(\text{Ti}_{0.9}\text{Sn}_{0.1})\text{O}_2$
 1000 °C $(\text{Ti}_{0.1}\text{Sn}_{0.9})\text{O}_2$

50/50 $\text{TiO}_2/\text{SnO}_2$



XRD-Phases

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 $(\text{Ti}_{0.2}\text{Sn}_{0.8})\text{O}_2$
 TiO_2
 Annealed – $(\text{Ti}_{0.2}\text{Sn}_{0.8})\text{O}_2$
 1000 °C $(\text{Ti}_{0.9}\text{Sn}_{0.1})\text{O}_2$

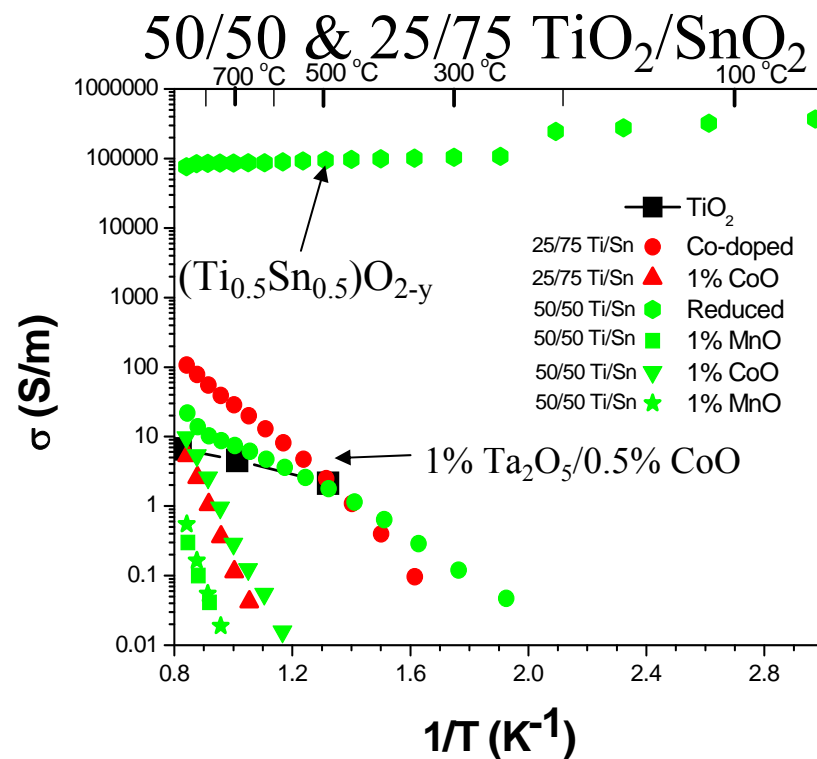
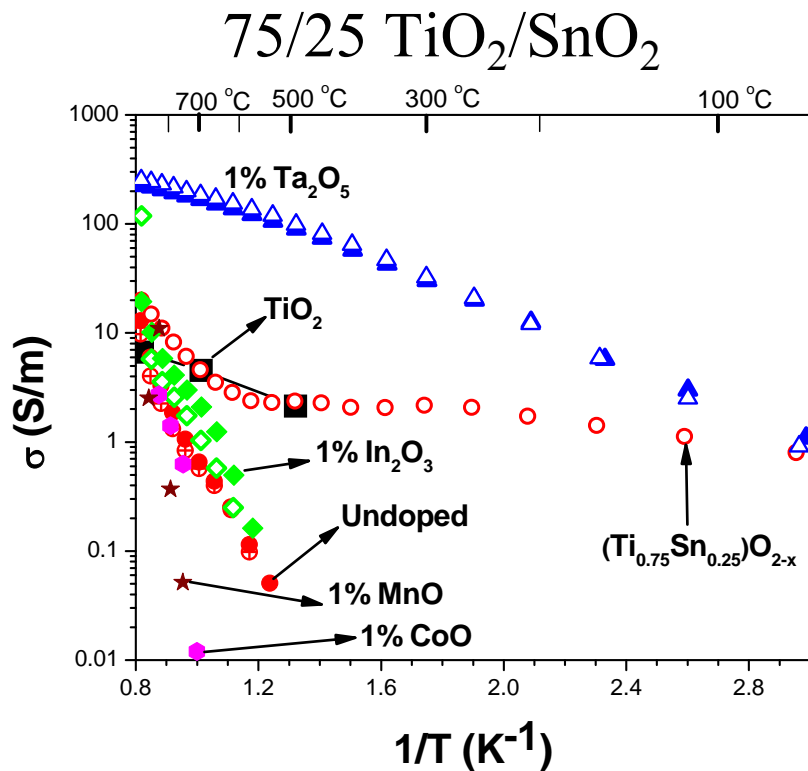
XRD-Phases

Sintered – $(\text{Ti}_{0.8}\text{Sn}_{0.2})\text{O}_2$
 $(\text{Ti}_{0.1}\text{Sn}_{0.9})\text{O}_2$
 Annealed – $(\text{Ti}_{0.2}\text{Sn}_{0.8})\text{O}_2$
 1000 °C $(\text{Ti}_{0.9}\text{Sn}_{0.1})\text{O}_2$

Microstructure
 Coarsening
 @ 1600 °C

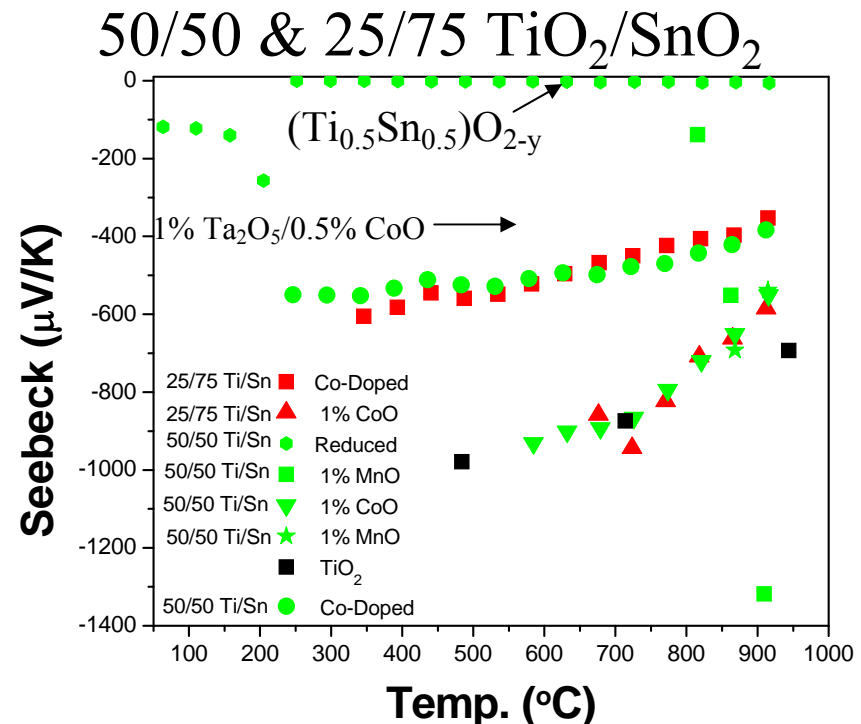
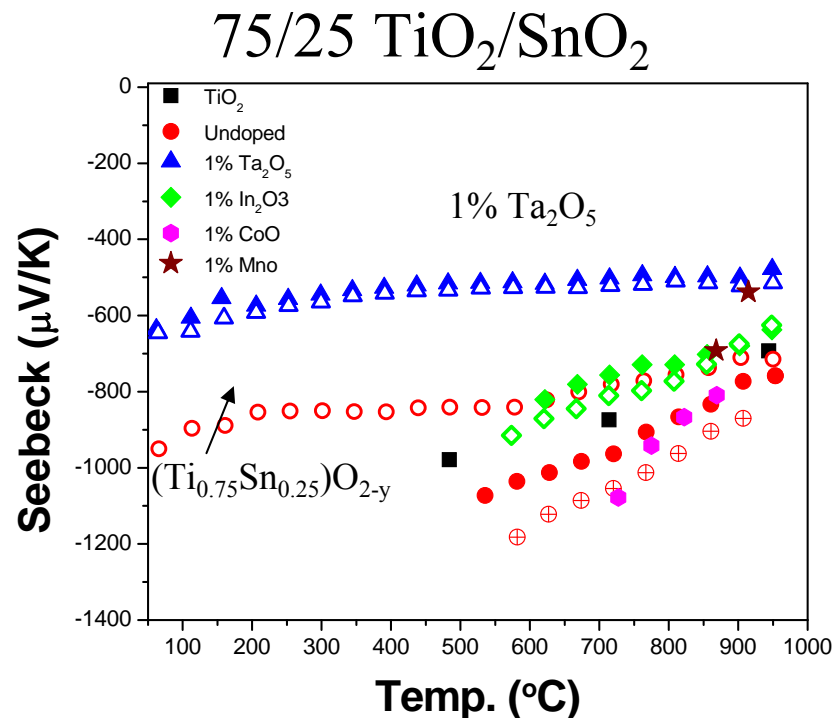
Grain Boundary Phases
 Segregation

Electrical Conductivity



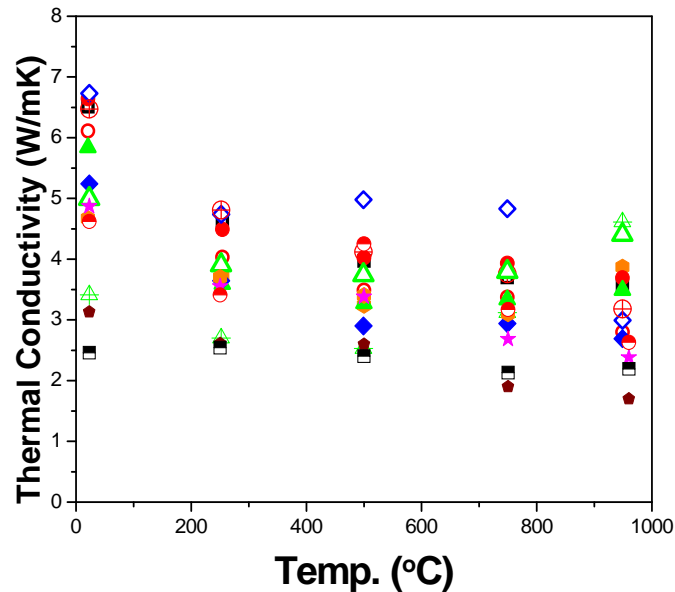
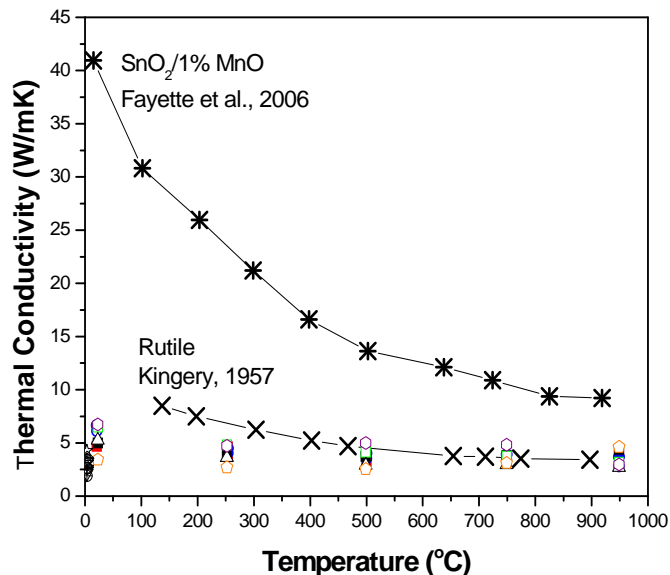
- Ta₂O₅ – Increases σ – $E_a \sim 0.25$ eV
- (Ti_xSn_{1-x})O_{2-y} – Oxygen Deficiency Increases σ – $E_a \sim 0.06$ eV
- Co-doping-Ta₂O₅/CoO - Increases σ – $E_a \sim 0.5-0.7$ eV
- In₂O₃, MnO & CoO – Ineffective in Enhancing σ – $E_a \sim 1-4.2$ eV

Seebeck Coefficient



- N-type
- Large Seebeck coefficients $> -400 \mu\text{V/K}$
- Large Seebeck coefficient – Low σ
- $(\text{Ti}_{0.5}\text{Sn}_{0.5})\text{O}_{2-y}$ low Seebeck ~ 0

Thermal Conductivity



Compositions

1% MnO-50 TiO₂
 1% CoO-50 TiO₂
 1% MnO-75 TiO₂
 1% CoO-75 TiO₂
 1% MnO-25 TiO₂
 1% CoO-25 TiO₂
 1% Ta₂O₅/0.5% CoO-25 TiO₂

- Compositions exhibit low κ – 1.7 to 6.8 W/mK
- Observe no dependence on composition or post treatments
- Spinodal Decomposition – κ reduction ?
- Best ZT ~ 0.05



In Summary

- $\text{TiO}_2/\text{SnO}_2$ compositions exhibit low thermal conductivity. Reduction in thermal conductance by spinodal microstructure has not been isolated.
- Improvements in electrical conductivity is needed. Grain boundary phases could be detrimental. Ta_2O_5 or oxygen deficiency enhances electrical conductivity.
- Sintering aids are required to densify equal-molar and tin oxide rich compositions. MnO and CoO promoted phase separation.